Study about the hydrogen embrittlement behavior of different highly alloyed austenitic alloys with different processing routes How different phases and crystallographic defects affect hydrogen embrittlement behavior? In the federal phase

H- assisted crack

2 µm ⁰ ¹⁰⁰ ²⁰⁰ ³⁰⁰ ⁴⁰⁰ ⁵⁰⁰ ⁶⁰⁰ ⁷⁰⁰

Temperature (°C)

Prof. Masoud Moshtaghi

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Steel Structures

Luther University

* Materials Science & Engineering A, Vol. 848 (2022) 143428. https://doi.org/10.1016/j.msea.2022.143428

0.00

0.05

Hydrogen desorption rate (wt.ppm s-1

0.30

LUT

H Hydrogen charging **H** H **1 μm** του 1919 και το 1919 initiation at δ/γ interface 1200 °C/h Peak 1 Peak 2012 - Peak 2013 Peak 3 (1991) and 3 (1992) Cumulative fit peaks of the peak phase **video Slip localization, and δ phase STEEL STRUCTURES**

σ

define the hydrogen embrittlement of the h

δ phase

behavior in Ni-Fe-Cr alloy.

Study about the hydrogen embrittlement behavior of different highly alloyed

so far? What is a pjo
is **Prospects of H2Pipeline projects in Austria What is done so far? What is understood?**

<u>masoud.moshtaghi@lut.fi</u> LUT University 0.15 Head of Steel Structures Research Group $-$ Masoud
Tenure T
Head of
_UT Univ
<u>masoud.</u> Tenure Track Assistant Professor Masoud Moshtaghi

2 A trailblazer in the field of design and strength analysis of metallic parts and components in Finland since 1974; towards testing and designing novel alloys and structures for energy and other structural applications.

Study about the hydrogen embrittlement behavior of different highly alloyed

austenitic alloys with different processing routes

Nickel alloys

Nickel alloys

Masoud Moshtaghi

Head of Steel Structures Research Group Tenure Track Assistant Professor

Mahdieh Safyari Post-doctoral researcher Hydrogen embrittlement and

additive manufacturing

Kalle
Post-o
Den **Kalle Lipiäinen** Post-doctoral researcher Mechanics of materials

Jani Riski PhD student Residual stress

Matti Koskimäki Research Engineer

+ 10 Bachelor students

+ 8 master students

+ 3 technicians

Hamidreza Rouhani

1.30 Finite element and machine learning, hydrogen-assisted Post-doctoral researcher fatigue

Antti Ahola Post-doctoral researcher Fatigue and stress analysis

POSI-doctorial researcher
Microstructure-mechanical Fatigue of welded aluminium **Dest-doctoral researcher** microstructure-mechanical
properties relationships **Shahriar Afkhami** Microstructure-mechanical

PhD Student
Hydrogen embrittlement initiation at θ Fatigue fatigue **Kiia Grönlund**

Study about the hydrogen embrittlement behavior of different highly alloyed

 \mathbf{L}

Donát Horváth PhD Student

Aleksi Härkönen PhD student Real-time fatigue monitoring

Pasi Tanskanen Lecturer

behavior in Ni-Fe-Cr alloy. **Tero Pesonen** PhD student Finite element methods for fatigue assessment

Sami Heinilä Lecturer

Edris Dabiri Lecturer (Metso)

Visiting Professor:

Prof. Martin Leitner From **Technical University of Graz, Austria**

Future new member:

Dr. Digvijay Singh (Starting from Jan. 2025) From **National Insitute of Materials Science (NIMS), Japan**

Overview of activities of LUT Steel Structures

- Design and testing hydrogen storage tanks, pipelines, etc.
- Hydrogen embrittlement and its mechanisms
- Steel structures for hydrogen energy
- Fatigue assessment methods and life prediction
- Static and fatigue strength of welded joints and components
- High Strength steels (HSS) and Ultra High Strength Steel (UHSS)
- High-cycle and low-cycle fatigue behaviour of components in view of their microstructure
- Microstructure-mechanical properties relationships
- Structural performance and quality of high-strength steels
- Enhancement of fatigue strength of welded joints by welding techniques and post-weld treatments
- Numerical methods and analysis of welded structures, incl. stress analysis and welding simulations
- Stability and distortion phenomena of thin-walled products
- Performance of steel structure at subzero temperatures
- Structural performance of AM components
- Failure analysis of structural components
- Environmental effect on mechanical response of components

Paper publication status

Paper contributions:

Number of publications in 2024

LUT Steel Structures in Scopus (2021-2024)

LUT Steel Structures in top 10 of the world:

Fatigue of high-strength steels

Hydrogen embrittlement

Key active projects:

- **Business Finland: FOSSA II – Fossil-free Steel Applications II**
- **Business Finland: CaNeLis - Carbon-neutral lightweight ship structures using advanced design, production, and life-cycle services**
- **Business Finland: AluWeld, Fatigue studies of welded aluminium structures**
- **Business Finland: Dreams, Database for Radically Enhancing Additive Manufacturing and Standardization**
- **Business Finland: Viima, Real-time fatigue studies of the machines**
- **FWF, Austrian Science Fund – ESPRIT Development of hydrogen-resistant high-strength steels**
- **HyGCEL project: Hydrogen and Carbon Value Chain in Finland**

+ more than 50 small-scale projects with national and international companies

Hydrogen embrittlement

Steel design for underground storage

Ship design and fatigue design

Design of hydrogen storage tanks under fatigue and vibration condition

Design of welded pipeline steels **Compressors** and gas turbines

Hydrogen pipeline projects lead by me in Austria

Stress intensity factor

A different approach to estimate whether an existing crack/flaw will grow is by looking at the stress intensity factor (SIF) at the crack tip. In your design classes you have already encountered the concept of stress rise due to stress concentrators.

Under the assumptions of LEFM we can derive the stress field in a cracked body, leading to:

$$
\sigma_{ij}=\bigg(\frac{k}{\sqrt{r}}\bigg)f_{ij}(\theta)+\sum_{m=0}^\infty A_M r^{m/2}g^{(m)}_{ij}(\theta)
$$

with r and θ being define at the orifin of the crack tip and counter-clockwise respectively. close to the crack tip, where $r\to 0$ the second term on the RHS of (8.1) vanishes. As a results, the $\sigma \propto r^{-1/2}$ relation holds in general for any cracked elastic body.

As you recall we can obtain the displacement directly from the stress field using the elastic constitutive equation to obtain the strains and then integrating them to obtain displacements. This implies that the displacements at the crack tip will be proportional to \sqrt{r} .

As we will see later, when deriving the crack tip fields, it is useful to replace k in (8.1) with $K = k\sqrt{2\pi}$.

Moreover, we will add a subscript to K to differentiate between the different crack opening modes: K_I ; K_{II} ; K_{III} .

 (5)

Stress intensity factor

When encountering a problem of mixed-mode, we can use superposition to find the stresses at the crack tip such that:

$$
\sigma_{ij}^{mixed}=\sigma_{ij}^{I}+\sigma_{ij}^{II}+\sigma_{ij}^{II}
$$

 (a)

Plain strain fracture

When we started discussing how to characterize a cracked structure we introduced the concept of a singularity dominated region and extracted a single parameter from it K which we claimed can fully characterize the crack tip fields.

The value of K at which a test specimen fails is denoted K_c and is a material parameter (i.e. independent on geometry).

- How can be concile the K approach to fracture arising from LEFM with the presence of plasticity at the crack tip we discussed last week?
- What are the conditions for which the K approach still holds?

Plain strain fracture

Cracks and flaws cause stress concentration

 $K_1 = Y \sigma \sqrt{\pi a}$

 K_1 - Stress intensity factor

 σ - Applied stress

- a edge crack length
- Y geometric constant

 K_{Ic} - critical value of stress intensity factor (fracture toughness)

Measuring Fracture Toughness: notch is machined in a specimen of thickness B $B \gg a$ plain strain condition. $B = 2.5(K_{1c}/Y$ ield strength $)^2$ Specimen is tensile tested Higher the K_{1c} value, more ductile the metal is

The plastic zone estimates we dealt with beofre are used in teh ASTM standards (e.e. ASTM E399) as a criteria for the validity of a K_I measurment.

Knowing the σ_y of our material, and after measuring the K_{Ic} from an experiment, we will use

$$
a,B,(W-a)\ge 2.5\left(\frac{K_{IC}}{\sigma_y}\right)
$$

Thickness, B

as a criteria for accepting our test result.

Mixed mode fracture

Mixed-mode scenarios will often be found in heterogenous structures (multi-phase materials, weldements, coatings, composites etc.).

We can use superposition and obtain the crack tip fields as

$$
\sigma_{11} = \frac{K_I}{\sqrt{2\pi r}} \left[\cos\frac{\theta}{2} \left(1 - \sin\frac{\theta}{2} \sin\frac{2\theta}{2} \right) \right] + \frac{K_{II}}{\sqrt{2\pi r}} \left[-\sin\frac{\theta}{2} \left(2 + \cos\frac{\theta}{2} \cos\frac{2\theta}{2} \right) \right]
$$

$$
\sigma_{22} = \frac{K_I}{\sqrt{2\pi r}} \left[\cos\frac{\theta}{2} \left(1 + \sin\frac{\theta}{2} \sin\frac{2\theta}{2} \right) \right] + \frac{K_{II}}{\sqrt{2\pi r}} \left[\sin\frac{\theta}{2} \cos\frac{\theta}{2} \cos\frac{2\theta}{2} \right]
$$

$$
\sigma_{12} = \frac{K_I}{\sqrt{2\pi r}} \left[\cos\frac{\theta}{2} \sin\frac{\theta}{2} \sin\frac{2\theta}{2} \right] + \frac{K_{II}}{\sqrt{2\pi r}} \left[\cos\frac{\theta}{2} \left(1 - \sin\frac{\theta}{2} \sin\frac{2\theta}{2} \right) \right]
$$

And the displacements:

$$
u_1 = \frac{K_I}{2\mu} \sqrt{\frac{r}{2\pi}} \left[\cos \frac{\theta}{2} (\kappa - \cos \theta) \right] + \frac{K_{II}}{2\mu} \sqrt{\frac{r}{2\pi}} \left[\sin \frac{\theta}{2} (\kappa - \cos \theta) \right]
$$

$$
u_2 = \frac{K_I}{2\mu} \sqrt{\frac{r}{2\pi}} \left[\sin \frac{\theta}{2} (\kappa - \cos \theta) \right] + \frac{K_{II}}{2\mu} \sqrt{\frac{r}{2\pi}} \left[-\cos \frac{\theta}{2} (\kappa - 2 + \cos \theta) \right]
$$

Elastic-Plastic Fracture Mechanics

J Integral

As you all remember we previously discussed energy methods in the context of fracture mechanics.

 G , the energy release rate was used when we discussed the stability of growing cracks.

The strain energy concept, proposed by Shih was used for defining the crack growth direction under mixed-mode conditions.

We will now introduce another energy related concept which will allow us to estimate the fracture toughness and resistance curves of cracks, while avoiding some of the limitations we discussed previously, resulting from the plastic zone size.

y L, B 51 -34.5 Bar N 1 Bar H $34.5 B$ The J integral is defined as : 0.6 0.4 0.2 Crack Extension, Aa $J = \int_{\Gamma} \left(W dy - \vec{T} \frac{\partial \vec{u}}{\partial x} ds \right)$

Elastic-Plastic Fracture Mechanics

CTOD - Crack Tip Opening Displacement

Using Irwin's crack-tip plasticity model, Wells was able to show that the CTOD (δ) follows:

$$
\delta = \frac{4}{\pi} \frac{G}{\sigma_y}
$$

A similar result is obtained following the strip model:

$$
\delta = \frac{G}{m\sigma_y}
$$

and m assumes a value of 1 for plane stress and 2 for plane strain.

While not entirely accurate, this assumption allows us to measure the Crack Mouth Opening Displacement (CMOD) and using similarity of triangles obtain the CTOD.

$$
\delta_{CTOD} = \frac{r(W-a)}{r(W-a)+a}\delta_{CMOL}
$$

The experimentaly measured CMOD is decomposed into an elastic and plastic part (similar to the way you would for a stress strain curve) and thus:

$$
\delta_{CTOD} = \delta_e + \delta_p = \frac{K_I^2}{m \sigma_y E'} + \frac{r(W-a)}{r(W-a)+a} \delta_{CMOD}
$$

Hydrogen pipeline test in Austria: measurement of toughness $\boldsymbol{\xi}$ **the value of the value**

ASME B31.12 features a nonlinear correlation (in logscale) between crack growth rate and SIF range. ASME B31.12 suggests a characteristic design da/dN curve for hydrogen pressure under 20 MPa (200 bar).

FEM and Machine Learning

Methodology Data Analytics

Application of Machine Learning

Development of predictive models

Actual value

- Assessment of various models
- Verification of developed model

MACHINE LEARNING

Predicted value Predicted value

LUT Steel Structures, Member of Standard Committee

❖ Prof. Masoud Moshtaghi is a member of the Standard Committee at NACE and API for hydrogen pipeline testing procedures.

GUIDE 21586 - Guidelines For Laboratory Testing For Hydrogen

Discussions 3 Libraries 0 Members 24

GUIDE 21586 - Guidelines For Laboratory Corrosion Testing For Hydrogen

last person joined 4 months ago

SC 26 - Carbon Capture, **Alternative Fuels, and Energy Storage**

SC 26 - Carbon Capture, Alternative Fuels, and Energy Storage

Hydrogen embrittlement in martensitic steels

Hydrogen embrittlement in steels

Martensitic steel design for steel pipelines in elastic loading regime in the high-pressure gaseous hydrogen condition

As-quenched + elastic straining in gaseous H [fractured]

Weld joints (SAW, FCAW and SMAW) of pipelines: towards hydrogen-resistant steel pipelines

2020/03/04 addition" International Journal of Hydrogen Energy, Vol. 47, 2022, 20676-20683. * M. Moshtaghi*, B. Loder, M. Safyari, T. Willidal, T. Hojo, G. Mori, "Hydrogen trapping and desorption affected by ferrite grain boundary types in shielded metal and flux-cored arc weldments with Ni

Hydrogen embrittlement in **ferritic and F/P steels**

Masoud Moshtaghi's Google Scholar: https://scholar.google.com/citations?user=UgjW2j8AAAAJ&hl=en

Activities and current projects: Hydrogen embrittlement in **pearlitic steels**

Masoud Moshtaghi's Google Scholar: <u>https://scholar.google.com/citations?user=UgjW2j8AAAAJ&hl=en</u>

Activities and current projects: Hydrogen embrittlement in **austenitic steels**

Masoud Moshtaghi's Google Scholar: https://scholar.google.com/citations?user=UgjW2j8AAAAJ&hl=en

Design and testing hydrogen pipelines with SSAB Tubulars

ars

SSAB
 $a_f < a_{cr} = \frac{K_c^2}{\pi(\sigma_{max}Y(a))}$
 $f = N_i + N_{cp} = N_i \int_{a_i}^{a_i} \frac{da}{C\Delta K(a)^m}$
 $f = \frac{1}{2} N_i + N_{cp} = N_i \int_{a_i}^{a_i} \frac{da}{C\Delta K(a)^m}$
 $f = \frac{1}{2} N_i + N_{cp} = N_i \int_{a_i}^{a_i} \frac{da}{C\Delta K(a)^{400}}$ **a SSAB**
 a $a_f < a_{cr} = \frac{K_{IC}^2}{\pi(\sigma_{max}Y(a))}$
 n.
 $N_f = N_i + N_{cp} = N_i \int_{a_i}^{a_f} \frac{da}{C\Delta K(a)^m}$
 give crack growth behaviour by are lastic fracture mechanics (LEFM) *C K a* ITS

SSAB
 $a_{\rm f} < a_{\rm cr} = \frac{K_{\rm IC}^2}{\pi (\sigma_{\rm max} Y(a))}$
 $= N_{\rm i} + N_{\rm cp} = N_{\rm i} \int_{a_{\rm i}}^{a_{\rm f}} \frac{da}{C \Delta K(a)^m}$

crack growth behaviour by

elastic fracture mechanics

(LEFM) **S**
 $f < a_{\text{cr}} = \frac{K}{\pi(\sigma_{\text{m}})}$
 $N_i + N_{\text{cp}} = N_i$ LUT

University
 $\frac{K_{\text{IC}}^2}{\sum_{\text{max}}^a Y(a)}$
 $\int_{a_i}^{a_i} \frac{da}{C\Delta K(a)^m}$

ehaviour by

mechanics **FS**
 SSAB
 $a_{\rm f} < a_{\rm cr} = \frac{K_{\rm IC}^2}{\pi(\sigma_{\rm max} Y(a))}$
 $= N_{\rm i} + N_{\rm cp} = N_{\rm i} \int_{a_{\rm i}}^{a_{\rm f}} \frac{da}{C\Delta K(a)}$
 $\text{crack growth behaviour}$ *Y* University
 *Y*_{IC}²
 *X*_{IC}²
 T($\sigma_{\text{max}} Y(a)$)
 *Y*₄ $\frac{da}{C\Delta K(a)^m}$
 *Y*₄ $\frac{da}{C\Delta K(a)^m}$
 *Y*₄ behaviour by
 *Y*ure mechanics S

S

S

S

S

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S

University
 $a_{cr} = \frac{K_{rc}^2}{\pi(\sigma_{max}Y(a))}$
 $N_i + N_{cp} = N_i \int_{a_i}^{a_i} \frac{da}{C\Delta K(a)^m}$

 $a_{\rm f}$ d_{α}

m

 $\int_{a_i} \frac{du}{C\Delta K(a)^m}$

 $(\sigma_{\max} Y(a))$

 $K_{\rm IC}^2$

2 IC

i

 $t = 4.2 \times 10^{-11} (AK)^{2.07}$

70 80 90 100

Fatigue crack growth behaviour by linear elastic fracture mechanics (LEFM)

50

 ΔK (MPa \sqrt{m})

60

 10^{-7}

 10^{-7}

 10^{-8} $\frac{1}{30}$

 $\frac{da}{dN}$ (m/cycle)

Fossil Free Steel Application (FOSSA) – Hydrogen-assisted fatigue in pipelines

Goal: Consideration of the effect of hydrogen exposure on fatigue design of Finnish pipelines Fatigue of a weld joint / welded component

Construction of a hydrogen exposure test device as part of the loading rig

Testing the steels and welded steels under the fatigue conditions in the presence of hydrogen. Analytical approach to the performance of the steels in fatigue conditions Ranking the materials based on their applicability in hydrogen-assisted fatigue applications. Studying the root cause failure of the specimens in the hydrogen-assisted fatigue condition.

Hydrogen-enhanced entropy (HEENT): a newly proposed mechanism for hydrogen embrittlement

Hydrogen-enhanced entropy (HEENT): A concept for hydrogen embrittlement prediction, International Journal of Hydrogen Energy, Volume 53, 2024, Pages 434-440.

HEENT mechanism: Makes the mechanism-based HEresistance alloy design and monitoring possible.

❖ The concept of hydrogen-enhanced entropy (HEENT) for hydrogen embrittlement prediction was introduced and discussed. The contribution of the different mechanisms to the total entropy, i.e. HEENT effect compensates for the total entropy reduction generated due to reduced fatigue life. This is applicable to the other types of hydrogenassisted fracture, based on HEDE, and HESIV, HELP, HELP+HEDE, etc.

- ❖ This makes feasible the mechanism-based design of HE resistant alloys and structures.
- ❖ With the monitoring the evolution of entropy and estimation of the generated entropy, one can reach to viable approach to estimate the efficiency of the different preventive actions for mitigating HE and also, indexing them. $4.0 - 1$

A value that is the property of the material irrespective of the hydrogen amount. This makes the prediction of the failure possible.

Hydrogen-enhanced entropy (HEENT): A concept for hydrogen embrittlement prediction, International Journal of Hydrogen Energy, Volume 53, 2024, Pages 434-440.

Hydrogen pipeline fatigue testing and analysis HyGCEL project

ASME B31.12 features a nonlinear correlation (in logscale) between crack growth rate and SIF range. ASME B31.12 suggests a characteristic design da/dN curve for hydrogen pressure under 20 MPa (200 bar).

$$
\frac{da}{dN}\!=\!a1\Delta K^{b1}+\left[\left(a2\Delta K^{b2}\right)^{-1}+\left(a3\Delta K^{b3}\right)^{-1}\right]^{-1}
$$

Design and testing hydrogen pipelines in Finland

Fatigue life as a function of pressure range for DN500 t ¼12.7 mm pipeline

Facilities developed so far in LUT

Facilities

Mechanical Testing and Fatigue Life evaluation

- High Cycle and Low Cyclic Fatigue Test with various test capacities
- Mechanical testing in different sizes and shapes
- Mechanical testing of the specimens in different environments
- Fracture mechanics testing approach, CT specimen
- High Cycle and Low Cyclic Testing
- Micro-Hardness Testing
- Slow Strain Rate Testing
- Fatigue testing in cryogenic condition
- Finite Element and Machine Learning

Fatigue crack growth behaviour by linear elastic fracture mechanics (LEFM)

MATERIAL TESTING MACHINES (LOAD FRAMES)

- . RUMUL Vibroforte 700 high frequency testing machine, April/2022
- 5 MN for static and dynamic loading
- 1200 kN and 750 kN for dynamic and static loading
- 400 kN for dynamic and static loading
- Hz1 and Hz2 frames for 150 kN dynamic and static loading
- 150 kN for dynamic and static loading
- 1 MN compression up to 7 m length columns and beams
- Drop weight testing machine for impact tests

Mechanical testing

(a) $\frac{da}{dN} = 1.1 \times 10^{-13} (\Delta K)^{4.00}$ $(m/cycle)$ $\frac{a}{b}$ $= 4.2 \times 10^{-11} (\Delta K)^{2.0}$ 60 70 80 ΔK (MPa \sqrt{m}) University

MATERIAL TESTING MACHINES (LOAD FRAMES)

- Laboratory have seven (7) servo hydraulic load frames for dynamic and static loading test set-ups.
- . Biggest test rig in Finland for dynamic testing up to 5 MN compression and tension loading.
	- Equipped with movable environment chamber down to -60°C to determinate material and connections behaviour at sub zero temperatures.
	- Full-scale tests of components made of high- and ultra-high-strength steels (S700-S1100).

Environment chamber + Cooling unit

Low-cycle fatigue assessment: experiment & simulation & LUT Equation Contrersity

Facilities

Facilities

Scanning electron microscopy (SEM) at LUT

Transmission electron microscopy (SEM) at LUT

Microstructural **Observation**

Microstructural Observation

- Optical Microscopy
- 3D Surface Measurement Device
- SEM/EDS/EBSD
- TEM
- Atom Probe Tomography

frame

MATERIAL TESTING MACHINES (LOAD FRAMES)

- Laboratory have seven (7) servo hydraulic load frames for dynamic and static loading test set-ups.
- Biggest test rig in Finland for dynamic testing up to 5 MN compression and tension loading.
	- Equipped with movable environment chamber down to -60°C to determinate material and connections behaviour at sub zero temperatures.
	- Full-scale tests of components made of high- and ultra-high-strength steels (S700-S1100).

Fatigue performace - SN-curves

Sub-zero fatigue testing

RHS X-joint (subzero)

Ultimate capacity

Ultimate capacity

Various postprocessing evaluation

15.0kV X2.00k SE

Zn $m - %$ Zinc inclusion to weld metal 100 µm

100 f

 $|0|$

LUT
Ce University

Fractograpghy

- SEM for analysis
- Connecting local quality and fatigue performance

Welding simulation

4R METHOD FOR FATIGUE DESIGN OF WELDED JOINTS AND COMPONENTS

LUT
Critique University

Novel multi-parametric fatigue assessment approach that considers:

- Material ultimate strength (R_m)
- Residual stresses (σ_{res})
- External stress ratio (R)
- Weld toe radius (r_{true})

Acting stress ratio (R_{local}) at notch root is obtained using well-known material models (Ramberg-Osgood and Neuber's notch theory) considering the four essential parameters

Symposiums organised by LUT Steel Structures

HRO SUUNNITTELUFOORUMI www.lut.fi/hro

Teräsrakenteiden laboratorion koordinoima foorumi vaativien rakenteiden suunnittelijoille, tuotekehittäjille ja tutkijoille sekä valmistuksesta, tarkastuksesta ja kunnossapidosta vastaaville

- SHY:n Suunnittelufoorumi
	- Yhteensä $38 +$ jäsenyritystä
	- Tavoitteet:
		- \checkmark Tuottaa uutta tutkimustietoa (tutkimusprojektit + HRO diplomityöt)
		- √Toteuttaa Suomen hitsaavan teollisuuden kannalta tärkeitä tutkimusprojekteja (BF, SA, EU..)
		- \checkmark Välittää uusin tarpeellinen tutkimustieto maailmalta kotimaiselle teollisuudelle (IIW)
		- √ Luoda ja ylläpitää alan yritysten välisten yhteistyötä ja kontaktointia
		- √Tuottaa palvelututkimusta ja koulutusta yrityksille (HRO alennus jäsenyrityksille)

HRO Suunnittelufoorumin teemapäivät

- \checkmark Alan viimeisimpien tutkimustulosten esittely (LUT + tutkimuslaitokset)
- √Jäsenyritysten ja kutsuvieraiden omat esitykset
- √Kansainvälinen vieraileva luennoitsija
- √Kaksipäiväiset vuosittain

Symposiums (co-)organised by LUT Steel Structures

The 24th European Conference on Fracture (ECF24) is organised by the European Structural Integrity Society (ESIS). ECF24 will be held on-site in Zagreb, Croatia, with possible online participation, from 26 - 30 August 2024.

HYDROGEN EMBRITTLEMENT IN METALLIC MATERIALS: PIPELINE TRANSPORT. **TC 21** HYDROGEN STORAGE, AND OTHER APPLICATIONS

https://ecf24.eu/symposia.html

TC21 "Hydrogen Embrittlement" ECF24 symposium organizers:

- Prof. Frank Cheng, FRSC, University of Calgary, Canada
- Prof. Tom Depover, University of Ghent, Belgium
- Prof. Milos Djukic, University of Belgrade, Serbia
- Prof. Motomichi Koyama, Tohoku University, Japan
- Prof. Livia Cupertino Malheiros, Imperial College London, UK
- Prof. Masoud Moshtaghi, LUT University, Finland
- Dr. Birhan Sefer, Swerim AB, Sweden

Commission XI: Pressure Vessels, Boilers and Pipelines & LUT Approxity

78th Annual Assembly and **International Conference**

June 22-27, 2025 - Genoa, Italy

Chair: Prof. Masoud Moshtaghi Head of Steel Structures RG, LUT University, Finland

Selected topics:

- **1- Welded hydrogen pipelines**
- **2- Wire arc additive manufacturing**
- **3- Laser welding**
- **4- Hybrid welding**
- **5- Laser powder bed fusion**
- **6- Hydrogen storage**
- **7- Hydrogen transport**
- **8- Underground hydrogen storage**

Commission XI Commission XI Pressure Vessels, Boilers and Pipelines

Contribution of LUT Steel Structures

6th Swedish Hydrogen seminar

- Embrittlement phenomena in metallic materials

Keynote speaker:

Prof. Masoud Moshtaghi

Design of Hydrogen Embrittlement Resistant High Strenght Steels for Different Applications

Date and time 10 October 2024 Location

Jernkontoret with online option Kungsträdgårdsgatan 10

Stockholm

Contact

Rachel Pettersson

rachel.pettersson@jernkontoret.se +46 8 679 17 04

<https://scholar.google.com/citations?user=UgjW2j8AAAAJ&hl=en> <https://www.scopus.com/authid/detail.uri?authorId=55387560300>

Thank you for your attention!

